(19)

Europäisches Patentamt

European Patent Office Office européen des brevets

(i) Publication number:

0 618 624 A2

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EUROPEAN PATENT APPLICATION

(1) Application number: 94104882.9

1 Int. Ci.5: H01L 33/00

2 Date of filing: 28.03.94

Priority: 29.03.93 JP 69938/93 18.03.94 JP 49022/94

Date of publication of application:
05.10.94 Bulletin 94/40

Designated Contracting-States:
AT BE CH DE DK ES FR GB GR IE IT LI LU MC
NL PT SE

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Light emitting device and method of manufacturing the same.

A light emitting device comprises a luminous region comprising a luminous porous material comprising a crystalline semiconductor and a non-porous region adjacent to the luminous region,

wherein a conductive type between both the regions is different at an interface between the luminous region and the non-porous region and the crystal structure between both the regions is continuous.

FIG. 1



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BACKGROUND OF THE INVENTION

Field of the Invention

The present invention relates to a light emitting device and a light writting device manufacturing

Related Background Art

Recently, porous materials have been remarkable as a new functional material used for an active structure factor in a device. The porous structure of IV-group semiconductor crystal, for example, has been chiefly used for the conventional SOt (Silicon On Insulator) structure forming process (refer to T. Unagami and M. Seki, J. Electrochem. Soc. 125, 1339 (1978)) because of its rapid oxidation capability. In this case, raw material has been used only as a structural material, in no connection with its electronic and optical properties. However, recently, a research and development has been vigorously made as for an application to light emitting devices which originates from a luminous phenomenon at room temperatures and with high efficiency (refer to L. T. Canham, Appl. Phys. Lett. 56, 1046 (1990)) because of the luminous phenomenon at low temperature (refer to C. Pickering, et al., J. Phys. C17, 6535 (1984)) due to a micro-structure with a desirable carrier quantum closing effect. Generally, the porous structure itself can be easily shaped by processing a raw material. If some problems are overcome to put into a practice use. applying functionally such porous materials may provide a very attractive new technology.

One largest obstacle which impedes putting the porous material to practical use is that it is difficult to inject current to the porous region. An active light emitting device must function as an electroluminescence (EL) device. In order to realize an EL device with higher efficiency, it is desirable to use a current injection-type device. It has been so far reported that a light emitting diode (LED) using a solid-state electrode on a current injection layer indicates far lower efficiency, in companson with the potential high photoluminescence (PL) efficiency of the same porous material. For example, a naked eye can clearly recognize under an illumination in a room that a porous silicon layer which is formed by anodizing the surface of a monocrystalline silicon substrate in a hydrolluoric acid solution is luminous over visible rays range in response to an illumination by a several-watt ultraviolet rays lamp so that the PL energy efficiency exceeds several %. However, since the current injection which is performed via a Schottky junction between a semi-transparent gold electrode and the same porous layer has less than a quantum elliciency of

10⁻³ % (reler to N. Koshida, et al., Appl. Phys. Lett. 60, 347 (1992)), a very high applied voltage of several hundred volts is needed to recognize a feeble luminescence with a naked eye in a dark place (refer to A. Richter, et al., IEEE Electron Device Lett. 12, 691 (1991)). An remarkable improvement is not seen in the LED including a conductive transparent electrode (refer to Namavar, et al., Appl. Phys. Lett. 60, 2514 (1992))

of an injection electrode ol indium tin oxide, an ntype micro-crystalline silicon carbide film (µc-SiC) (refer to T. Futagi, et al., Jpn. J. Appl. Phys. 31, L616 (1992)), and a pn-heterojunction, in order to overcome such a situation. In the example where an n-type gallium phosphorus (GaP) is used for the same purpose, even a current injection has not en yet succeeded in an actual case (refer to J. C. Campbell, et al., Appl. Phys. 60, 889 (1992)).

On the other hand, it has been reported that the current injection in which a porous silicon layer is immersed in an electrolytic solution used for an anodization to utilize the solid-liquid interface in the internal walls in pores of the porous silicon improves the luminous efficiency, compared with the above solid-state electrode (refer to A. Halimaoui et al., Appl. Phys. Lett. 59, 304 (1991), P. M. M. C. Bressers, et al., Appl., Phys., Lett, 61, 108(1992). E. Bustarret, et al., Appl. Phys. Lett. 61, 1552-(1992), L. T. Canham, et al., Appl. Phys. Lett. 61, 2563 (1992). The above method is poor in practical use because the porous silicon layer is etched at a light emission so that the light emission disappears soon. However, it is noted that there is a possibility that the current injection efficiency dominates largely the injection-type EL efficiency of porous silicon. From this viewpoint, it is considered that the poor interlace between the electrode and the luminous porous silicon layer impedes the current injection, thus causing the poor emitting efficiency of the solid-state injection electrode type LED. Therefore, it is desirable to introduce a new injection electrode material with good interface property over the above heterojunction example.

SUMMARY OF THE INVENTION

In order to overcome the above mentioned technical problems, an object of the present invention is to provide a light emitting device which has a reduced contact resistance between its electrode and its furninous region.

Another object of the present invention is to provide a light emitting device which can perform a current injection with high efficiency.

Further another object of the present invention is to provide a light emitting device having an excellent luminous efficiency.

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BRIEF DESCRIPTION OF THE DRAWINGS

Fig. 1 is a schematic diagram showing a structural embodiment of the light emitting device according to the present invention;

Figs. 2A to 2D are schematic diagrams each showing an embodiment of the light emitting device according to the present invention; Figs. 3A to 3D are schematic diagrams each

showing an embodiment of the light emitting device according to the present invention; Figs. 4A to 4C are schematic diagrams showing an embodiment of a light emitting device manu-

facturing process according to the present invention; Figs. 5A to 5C are schematic diagrams showing

Figs. 5A to 5C are schematic diagrams showing an embodiment of a light emitting device manufacturing process according to the present invention:

Fig. 6 is a schematic dagram showing an embodiment of a light emitting device manufacturing process according to the present invention. Fig. 7 is a schematic diagram showing an embodiment of a light emitting device manufacturing process according to the present invention; Figs. 84 to 82 or schematic diagrams showing an embodiment of a light emitting device manufacturing process according to the present in-

Figs. 9A to 9C are schematic diagrams showing an embodiment of a light emitting device manufacturing process according to the present invention:

Figs. 10A to 10D are schematic diagrams showing an embodiment of a light emitting device manufacturing process according to the present invention:

Fig. 11 is a schematic diagram showing an embodiment of a light emitting device manufacturing process according to the present invention:

Figs. 12A to 12D are schematic diagrams showing an embodiment of a light emitting device manufacturing process according to the present invention;

3

Fig. 13 is a schematic diagram showing an embodiment of a light emitting device manufacturing process according to the present invention:

Figs. 14A to 14F are schematic diagrams showing an embodiment of a light emitting device manufacturing process according to the present invention;

Figs. 15A to 15D are schematic diagrams showing an embodiment of a light emitting device manufacturing process according to tee present invention:

Figs. 16A to 16E are schematic diagrams showing an embodiment of a light emitting device manufacturing process according to the present invention; and

Figs. 17A to 17D are model diagrams showing an embodiment of a light emitting device manufacturing process according to the present invention.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

The present invention results from that the present inventors have zealously studied to overcome the above problems.

The light emitting device according to the present invention is characterized by a uninous region which comprises a luminous procus material comprising a crystaline semiconoluctor and a nonprocus region adjacent to the luminous region wherein the luminous region and the non-procus region have a different conductive type to each other at the interface thereof and the crystal structure between both the regions is continuous.

The first preferred embodiment of the present invention is a method of manufacturing a sligh emitting device which is characterized by the steps of forming a member having a non-prous crystal region and a porcus crystal region of a conductivity pipe different from that of the non-prous crystal region, so as to continue crystal structure of a crystal region of the porcus crystal region of the porcus crystal region and structure of the province syntax incursor of the porcus crystal region and the province crystal region and the province crystal region of the porcus crystal region and the province crystal region of the province crystal region of

The second preferred embodiment of the present invention is a method of manufacturing as legist emitting device which is characterized by the steps of ferming a member having a non-process crystal region and a procus crystal region so as to continue a crystal structure of the non-process process crystal structure of the non-process process crystal region and a crystal structure of the non-process process crystal region and forming a luminous region in the procus crystal region.

EP

According to the present invention, the light emitting device has an element structure where non-porous region with a continuous crystal structure is arranged to be adjacent to a luminous region comprising a porous material and the nonporous region acting" at an electrode for injecting current to the luminous region, so that the perfect bonding between the electrode and the luminous region can reduce its contact resistance. The current can be injected from the junction region with high efficiency by making the conductivity type of the non-porous region different from that of the luminous region. As a result, a light emitting device can be provided which has a good luminous efficiency and is practical as an entire element.

An explanation will be made below in detail as for the light emitting device and the light emitting device manufacturing emethod according to the present invention, with reference to the drawings.

Fig. 1 is a schematic cross sectional view showing the most conceptual element structure of an light emitting device according to the present invention. The light emitting device has a structure where the luminous region 2 comprising a luminous porous material is sandwiched between the current injection electrode regions 1, 3. The injection electrode region 1 is formed of a non-porous material with a low resistance which is same as the mother material for the luminous region 2. The interface 4 between both the regions has a continuous crystal structure. Both the regions have a different conductive type from each other, and homogeneous pn-type junction is formed there. The injection electrode region 3 does not depend on its structure if it is of a low resistance. The injection electrode region 3 has the same conductive type as that of the luminous region 2. The interface between both the regions has a continuous crystal structure. When a DC current is conducted between the injection electrode regions 1 and 3 via the extracting electrode 5 and 5' connected to the device, electric charges are injected from the homogeneous pn-junction interface 4 to the luminous region 2 so that fight emission occurs. When the extracting electrode 5' is directly bonded to the light emitting region 2 with a low contact resistance, the injection electrode region 3 can be omitted.

The point of the present invention will be described by studying how the interface property affects the current injection efficiency of the entire element. First, (1) the good ohmic contact between the extracting metal contact 5 and the injection electrode region 1 is confirmed because the injection electrode region 1 is formed of a non-porous material with low resistance. (2) A good homogeneous pn-junction can be formed because the crystal structure continues at the junction interface

4 between the injection electrode region 1 and the luminous region 2 of the same material and along both the regions sharing a common mother material. As described in Related Background Art, the interface characteristics dominate current injection efficiency and the luminous efficiency of the element. Hence, the light emitting device according to the present invention has a high efficiency by forming an injection electrode region which is in "non porous structure", which has "a crystal structure" continuous to the luminous region 2, and which is of "the same material" and of "heterojunctiontype". Moreover, when the injection electrode region 3 is provided, (3) the contact between the luminous region 2 and the injection electrode region 3 is no matter because of "continuity of crystal structure", and (4) it is expected that the good ohmic contact between the injection electrode region 3 and the extracting electrode 5' results from the low resistivity of the injection electrode region

It is not necessarily needed that the luminous region 2 is made of an uniform furninous porous material along the whole region. The luminous region may be formed of plural porous or non-porous regions of different structure. It is required that at least one of the regions must be a luminous material. Therefore, if other regions are offered as other uses except the luminous function, both the device characteristics and the device manufacture ing method may be convenient in some cases. For example, the vicinity of the interface of the injection electrode region 1 is made non-porous or nonluminous porous material having relatively large residual structure, whereby the pn-junction can further be improved at the interface 4. Of the light emitting device manufacturing methods according to the present invention (to be described later), the method which uses an epitaxial growth can further improve the epitaxial interface and the film quality

of an epitaxial layer. In the porous material existing in the light emitting region 2, the structure is formed of the same mother material as the injection electrode region 1 or 3, whereby it is not important that a heteromaterial may be formed on the surfaces facing pores in the residual structure. Moreover, all spaces in a pore may be filled with a heteromaterial. Generally, in some cases, the mechanical and thermal stability is not enough because the luminous porous material has a very fine residual structure. Hence, in some cases, an effective reinforcement may be made using the above heteromaterial

Figs. 2A to 2D illustrate a spacial arrangement of a luminous region comprising a non-porous region and a luminous porous material as a more concrete embodiment of a light emitting device

Figs. 2A to 2D show an example where a current route is vertically formed on the surface of a monocrystalline substrate in a water form. In the most simplified device structure, shown in Fig. 2A, a pn-junction is thinhed between the non-porous layer t of a first conductive type and the tuminous porous layer 2 of a second conductive type, with a continuous crystal structure in the pn-junction. The extracting electrodes 5 and 5' are provided on the ends of the element structure, respectively. Like numerals represents regions corresponding to those in Fig. 1. Depending on the interface characteristics, the intermediate layer 2' can be arranged between the nonporous layer t and the luminous porous layer 2 (Fig. 2B). This example shows that the luminous region 2 includes an region other than the luminous porous region in the structure shown in Fig. t. That is, the intermediate layer 2' includes the luminous region 2 shown in Fig. 1. The intermediate region 2' may be formed of, for example, a non-luminous porous material or a non-porous material including monocrystal. However, the conductive type of the intermediate 2' should be same as that of the luminous porous layer 2, and the layers on the front and back of the intermediate layer 2' should be continued in crystal structure. However, it is not necessarily needed that the interface between the intermediate layer 2' and other layer in the luminous region 2 is sharp but it may be varied restrictively and continuously. Depending on the characteristics of the interface between the extracting electrode 5' and the luminous porous layer 2, additional injection electrode layer 3 may be provided (Fig. 2C). This is an example showing the injection electrode 3 which is provided in the device structure shown in Fig. 1. The injection electrode layer 3 is formed of, for example, a non-luminous porous material or nonporous material including monocrystal. However, the conductive type of the injection electrode layer 3 should be same as the luminous non-porous layer 2. The crystal structure between the injection electrode tayer 3 and the luminous non-porous electrode layer 2 should be continued in crystal structure. However, the interface is not necessarily needed to be sharp, but may restrictively and continuously vary. Fig. 2D illustrates an example in which the intermediate layer 2' serving also as the

Figs. 3A to 30 are an embodiment showing that an electrode route is formed in a direction parallel to a surface of noncrystal film provided on the insulating surface of, for example, the substrate in water form 0. Each structural element and its function correspond to those shown in Figs. 2A to 2D. The extracting electrodes 5 and 5° are in contact with the film surface. However, the geometric

injection electrode layer 3.

ric arrangement should not be limited to the

Next, as for the light emitting device manufacturing method according to the present invention, the device of an embodiment shown in Figs. 24 to 2D and 3A to 3D will be explained in accordance with a process for forming a light emitting region comprising a luminous porous material and a nonprosous region each having a different conductive

type, so as to form a continuous crystal structure. The first method of manufacturing a device with the structure shown in Fig. 2A is shown in Figs. 4A to 4C. First, a non-porous substrate 1 having a desired conductive type and a resistance is prepared (Fig. 4A). The layer 20, which has a conductive type different from that of the substrate 1 but has the same mother material and has a continuous structure at the interface 4, is provided on the surface of the substrate 1 (Fig. 4B). In a concrete method, the layer 20 may be formed by epitaxial growth on the surface of the substrate 1 with impurity elements being introduced to control the conductive type, or the layer 20 is internally formed from the surface of the substrate 1 by effecting counter-doping by an ion implantation from the surface of the substrate t, a solid-phase ditfusion from a deposited, or a diffusion from a gas-phase film. Moreover, it is possible to combine the above methods. Next the luminous region 2 comprising a luminous porous material can be made by making the layer 20 porous from the surface to the interface 4 by using a method of converting a non-porous material into a porous material, such as anodization or a photo-formation (Fig. 4C). Next, the extracting electrodes 5 and 5 are formed. Finally, a device with the structure shown in Fig. 2A can be formed. Since the pnjunction interface 4 has a continuous crystal structure before a porous structure formation, the origi nal state is maintained even after a final step.

Next, the second method of manufacturing a device with the structure shown in Fig. 2A is shown in Figs. 5A to 5C. First, a non-porous substrate 1 with a desired conductive type and a desired resistance is prepared (Fig. 5A). A porous structure formation is effected from the surface of the substrate 1 to a desired depth (Fig. 5B). Next, a counter doping is carried out from the surface of the porous structure formation layer 20 to make the luminous porous layer 2 having a different conductive type from that of the substrate 1 (Fig. 5C) Then a device with the structure shown in Fig. 2A can be formed by forming the extracting electrodes 5 and 5'. In the same manner as those in Figs. 4A to 4D, the crystal structure, needless to say, is continuous at the pn-junction interface 4.

Methods of manufacturing the device shown in Fig. 2B will be described as follows. The first FP

methods can be performed by slightly modifying the methods each which forms the device shown in Fig. 2A explained by referring to Figs. 4A to 4C and 5A to 5C. First, in the step as shown in Fig. 4B, the intermediate layer 2' is left to a desired thickness by terminating porous structure formation just before the formation reaches the interface 4 from the surface of the fayer 2 (Fig 6). In the step shown in Fig. 4C, a counter doping is effected over the surface of the porous structure formation layer 20 to the non-porous region of the substrate 1 to form the intermediate layer 2' having a desired thickness (Fig. 7). In either case, the intermediate layer 2' is made of a non-porous material of the same conductive type as that of the luminous porous layer 2 and has a continuous crystal structure at the pn-junction interface 4 to the substrate 1.

The second method of manufacturing the de vice shown in Fig. 2B is not applied to the step in which a pn-junction is formed by counter-doping after the porous structure layer is formed as shown in Figs. 5A to 5C. The reason is that the obenomenon is utilized, that both the porous structure formation and the porous material structure are largely influenced by the composition of non-porous material, impurity concentration, and other factors. For example, in the porous silicon formation by anodization, the structure strength increases because when a heavy impurity concentration in a silicon substrate lowers a resistance the residual structure is large and the porous degree lowers. thus suppressing a resistance increase due to porous material. In this case, a non-luminous porous material is formed. In the steps explained with reference to Figs. 8A to 8C, a non-luminous porous material of low resistance is used as the intermediate layer 2' shown in Fig. 2B. First, the non-porous substrate 1 having a desired conductive type and a resistance is prepared (Fig. 8A). On a surface of the substrate, the layer 20', which has conductive type different from that of the substrate 1 but has the same mother material and has a continuous crystal structure at the interface 4, is provided. Further the layer 20, which has the same conductive type as that of layer 20' but has a different impurity concentration and has a continuous crystal structure (Fig. 8B). For example, the impurity concentration of the layer 20' may be set to a higher amount than that of the tayer 20. The process used in the step shown in Fig. 4B can be used in the concrete method of manufacturing the layers 20 and 20'. For example, when a formation is effected by epitaxial growth, layer 20 may be deposited after a deposition of the layer 20°. In ion implanta tion, multiple step ion-implantation with varying implanting energy conditions or ion species, or an ion-implanting profile may be used, whereby the spatial transition between the layers 20 and 20'

becomes continuous. Next, by making the layers 20 and 20' porous from the surface to the interface 4 (Fig. 8C), the luminous region 2 comprising a luminous porous material, the layer 20' acting as the intermediate layer 2', is formed. Then the extracting electrodes 5, 5' are formed on the luminous region 2 to form a light emitting device with the structure shown in Fig. 2B. In this case, the crystal structure at the pn-junction interface is continuous

10

Next, the first method of manufacturing a device with the structure shown in Fig. 2C is shown in Figs. 9A to 9C. First, the non-porous substrate 1 having a desired conductive type and a resistance is prepared (Fig. 9A). On a surface of the substrate. the layer 20, which has a conductive type different from that of the substrate 1 but has the same mother material and has a continuous crystal structure at the interface 4, is provided. Further, the layer 30, which has the same conductive type as that of the layer 20 and has a different impurity concentration, is provided (Fig. 9B). For example, the impurity concentration of the layer 30 may be higher than that of the layer 20. The process used in the step shown in Fig. 4B can be used as the concrete method of manufacturing the layers 20 and 30. For example, the formation is effected by epitaxial growth, the layer 30 is deposited after a deposition of the layer 20. In ion implantation, multiple step ion-implantation with varying implanting energy conditions or ion species, or an ionimplanting profile may be used, whereby the spa tial transition between the layers 20 and 30 becomes continuous. Next, by making the tayers 30 and 20 porous from the surface to the interface 4 (Fig. 9C), both the layers become an injection electrode layer 3 of a non-luminous porous material and the luminous layer 2 comprising a luminous porous material. Then the extracting electrodes 5 and 5' are formed on the luminous layer 2 to form a light emitting device with the structure shown in Fig. 2B. In this case, the crystal structure at the pnjunction interface 4 is continuous. The second method of manufacturing a device

with the structure shown in Fig. 2C is shown in Figs. 10A to 10D. First, the non-porous substrate 1 having a desired conductive type and a resistance is prepared (Fig. 10A). On a surface of the substrate, the layer 20, which has a conductive type different from that of the substrate 1 but has the same mother material and has a continuous crystal structure at the interface 4, is provided (Fig. 10B). The concrete method of manufacturing the layer 20 is the same as the process described above. Next the layer 20 is made porous from the surface to the interface 4 to form the luminous layer 2 comprising a luminous porous material (Fig. t0C). The injection electrode layer 3 of a non-porous material is deposited on the surface by epitaxial growth etc. (Fig. 10D). Since the injection electrode layer 3 should be of the same conductive type as that of the light emitting depice 2 and have a low resistance, the impurities are introduced at a deposition or are doped in a 'faith' store.

By performing both the first and second methods for forming a device with the structure shown in Fig. 2C, it is possible to form the injection electrode layer 3 of the plural layers shown in Fig. 11. In this case, the non-process injection electrode layer 32 may be deposited by epitaxial growth etc. on the non-luminous persus injection electrode layer 31 formed according to the first method.

In the third method for manufacturing a device shown in Fig. 2C, the order of the layer manufacturing steps described above is reversed. The third process will be explained below with reference to Figs. 12A to 12D. First, the non-porous substrate 0 having a desired conductive type and a resistance is prepared (Fig. 12A). The substrate 0 is made porous to a desired depth from the surface thereof to form the luminous layer 2 comprising a luminous porous material (Fig. 12B). The injection electrode layer 1 is deposited by epitaxial growth etc. on the surface of the substrate 0 (Fig. 12C). In this case, the injection electrode layer 1 should be of a nonporous material with a low resistance. The interface 4 between the injection electrode layer 1 and the luminous layer 2 should be a continuous crystal structure and the substrate 0 should be different from the luminous layer 2 in a conductive type. Therefore, it is required that impurities are introduced at a deposition or are doped in a later step. Putting the substrate upside down makes the structure shown in Fig. 12D. The pn-junction interface 4 is formed between the injection electrode layer t finally deposited and the luminous layer 2. The injection electrode fayer 3 which has the same conductivity as that of the luminous layer 2 in the substrate 0 left without being made porous is formed on the luminous layer 2. The extracting electrodes 5 and 5' are formed to obtain a device shown in Fig. 2C.

The device shown in Fig. 20 can be tormed by combining the method of manufacturing the device shown in Fig. 28 flar and the method of manufacturing the device shown in Fig. 28. For example, referring to Fig. 2A, after a tayer which is of the same conductive type as that of the substrate? and has a high impurity concentration is formed on the surface of the morphorous substrate of, the process goes to the stops shown in Fig. 128. Thus the high impurity concentration bayer is convented to the intermedate layer 2° of non-huminous protus maintained by the substrate shown in Fig. 13 is obtained and the device shown in Fig. 20 can be formed.

7

The method of manufacturing the device group shown in Figs. 3A to 3D is the same as that shown in Figs. 2A to 2D except that a non-porous film on the insulating surface of a substrate is made porous. An explanation will be made as for some typical forming steps with reference to the device structure in Fig. 3D as an example.

Figs. 14A to 14F show the first method of manufacturing the device in Fig. 3D. First, a substrate on which a non-norous film 20 is formed on the insulation surface of the substrate 0 is provided (Fig. 14A). Next, using a photolithography process and a local doping process such as a focused ion beam technique which are conventionally used in the semiconductor integrated circuit fabrication, the region 10 having a conductive type different from that of the non-porous fitm 20 is formed in the nonporous film 20 and the region 30 having the same conductive type as that of the non-porous film 20 and having a high impurity concentration and a high resistance is formed (Fig. 14B). The take-out electrodes 5 and 5' in contact with the regions 10 and 30, respectively, are formed using the patterning technique (Fig. 14C). Furthermore, a protective layer 6 is formed over the electrodes 5 and 5' (Fig. 14D). Since the protective layer 6 protects the structure underneath it in the following step when the region 20 of the remaining non-porous film is subjected to a porous structure forming process, it must sufficiently withstand against the porous structure forming process. The protective film 6 on the left end of the opening is offset from the boundary between the regions 10 and 20 as shown in the drawing. Next, the region 20 is converted into a porous layer. In the case where an anodizing process is used as a concrete porous structure forming process, the entire substrate is immersed in an electrolytic solution. Current is conducted between the take-out electrode 5' and the counter electrode 7 (Fig. 14E). In the case of a photo formation, the porous structure forming process is carried out merely by illuminating uniform rays, in the above steps, of the region 20, the region underneath the opening in the protective film 6 is converted into the turninous region 2 including a fuminous porous material, the offset part is converted into the intermediate layer 2', and the regions 10 and 30 respectively converted into the injection electrode regions 1 and 3. The device shown in Fig. 3D having a pn-junction at the interface 4 with a continuous crystal structure is formed (Fig. 14F), If the offset of the protective tilm 6 at the opening is omitted, the device shown in Fig. 3C is formed It formation of the region 30 is omitted, the device shown in Fig. 3B is formed. If the offset at the opening in the protective film 6 and lormation of the region 30 are omitted, the device shown in Fig. 3A is formed.

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steps.

A detailed explanation will be made below as for the examples of a light emitting device and a light emitting device manufacturing method according to the present invention with reference to examples using a silicon crystalline material.

(Example 1)

The example of a light emitting device having the structure shown in Fig. 2A shown with the forming steps in Figs. 4A to 4C will be explained

First, a phosphous-dopoud n-type slittorn moncorystalline water which has a face orientation of corystalline water which has a face orientation of (100 and a resistivity of 10.02 e-cm is prepared. On the surface of the water, a p-type monocrystalline siltion layer of a thickness of about 1 sum was epitaxially grown using a CVD process simultaneously using dichlorosilane pas and dishorane gas. Auminum of a limbickness of 2000A was vapordeposited on the back surface of the water to ansure confect officially confections.

Next, only the water surface was contacted with a hydrofluoric acid ethanol solution of 20 wrfs so as to face e platinum plane electrode. While the water surface was lifuminated with a 1 kW halogen itamp, a DC voltage was applied between the water acting as an anode electrode and the platinum electrode. An anodization was performed for two

minutes while maintaining the current density on the wafer surface to 10 mA+cm-2. When the cross section of a sample particularly prepared to confirm the porous structure formation was observed under an electron microscope with a high resolving power, it was confirmed that the anodization advanced just to a depth of 1 µm from the wafer surface to form a porous layer and the residual structure of the porous layer was perfectly continued to the crystal structure of the substrate. This means that the entire p-type epitaxially grown Si layer was converted into a porous layer. Illuminating the surface of the porous layer with a 5 watt ultraviolet lamp gave a red PL. Since the luminous porous layer had a very fine and brittle structure, only the surface of the residual structure was oxidized by a RTO (Rapid Thermal Oxidation) process to stabilize it. By this process, the PL emission peak was somewhet blue-shifted and the strength was increased about five times.

14

After it was confirmed that the procus layer oxidized and having an increased Pt-efficiency was sufficiently in a stable state, an SiO₂ tilm of a trickness of 2000Å was again deposited on the water surface by using the CVD process. Then some 2 mm square openings were formed in the oxide film by the conventional phototilhography process. In terming the openings, the oxide film on the top surface of the oxidized porous layer as the underlying layer must also be removed. Before a re-formation of a natural oxide film in the opening, a semi-irans-pant pdd thin film of a film thickness of 100Å was obsposited thereto. Moreover, an At-

the contacts on the front and the back surfaces of the wafer of the device prepared by the above process, a rectification characteristic was indicated where the current direction with the front surfece contact side being the positive electrode is the forward direction, and a visible area luminescence of nearly orange color occurred from the openings in the oxide film at a threshold voltage of about 3 volts. It is assumed that this luminescence is an electroluminescence due to a current injection into the porous silicon layer with an ultra-fine structure via the junction interface between the n-type monocrystalline layer of the substrate and the p-type porous layer just thereon. It is assumed that the reason for the luminous threshold voltage is practically and sufficiently low is that the continuous crystal structure in the pn-junction interface provides good rectification characteristic whereby the current injection efficiency is higher than thet of a conventional structure

(Example 2)

An explanation will be made as for an example of a light emitting bevice with the structure shown in Fig. 2B manufactured in the forming steps shown in Fig. 6. or 15

When a CC voltage was applied between the contects on the form and back surfaces of a wafer, the device showed a rectification ratio which is higher by nearly one order than that of the device of Example 1. The threshold voltage increased sightly a book 35 ords but al unimous efficiency was obtained which was several times stronger han that of the device of Example 1. It is presumed that the improved rectification characteristic results from that the fort and back points of the pre-private results from that the fort and back points of the pre-private results from that the efficiency of current injection into the furnious prorus layer, thus increasing the furninous efficiency.

(Example 3)

An explanation will be made as for an example of a light emitting device having the structure shown in Fig. 2B manufactured in the steps shown in Figs 84 to 8C.

Ā device was prepared by ripositing the device preparation risps of Example 1 with the exception that the step of growing a p-type epitisais layer on an -type monorsystaline substate was divided into two steps. The two steps for forming a p-type special content of the steps of the step of the special content of the steps of the step of should be step of the step of the step of discrete gas increased that limites and then reducing the discrete size of the step of steps of the step of the step of the step of the step of properties of the step of steps of the step of steps of

The observation of the cross section of a sample found that the anodization proceeds from the wafer surface to about 1 µm depth to form a porcous layer. However, according to a further detail observation, it was confirmed that a porcous material

having ultra-line pores was formed to a depth of about 0.8 µm from the water surface and the layer of a thickness of about 0.2 µm under the ultra-line porous material was formed of a porous material to large pores having an average structure larger by about two figures in structure.

Conducting a DC current between the electrodes on the front and back surfaces of the water found that the device had the intermediate charactinistic between the devices according to Example 1 and 2 in items including the rectification characteristics, luminous threshold voltage and luminous efficiency. In this case, its presumed that the propus layer with large peres functions as an intermediate layer.

(Example 4)

An explanation will be made as for an example of a light emitting device having the structure shown in Fig. 2A manufactured in the steps in Figs. 5A to 5C

First, a boron-doped p-type silicon morcrystaline water having a face orientation of 1000 and a resistivity of 10 p-cm was prepared. Boron ions were accelerated at 50 keV to be implanted into the back surface of the water with a dose of 5 x 10th cm⁻². Then the water was subjected to a thermal ameasing at 950°C for 30 minutes in nitrogen atmosphere to be activater to be

Next, the substrate was arranged between a pair of parallel platinum plane electrodes and immersed in a hydrofluoric acid-ethanol solution of a concentration of 25 wt%. Anodization was performed by applying a DC voltage between one platinum electrode acting as an anode electrode facing the back surface of the wafer and the other platinum electrode for one minute, with the region other than the wafer surface being electrically insulated, while the current density on the wafer surface was controlled to 10 mA+cm-2. Then, the wafer was left for 10 minutes under a lighting in a room, with the active circuit electrically short-circuited. When the cross section of an observation sample prepared particularly to check the porous forming process was observed under a high resolution electron microscope, it was confirmed that the anodization advanced from the surface of the wafer just to 0.5 µm deep to form a porous layer. When the surface of the porous layer were irradiated with a 5-watt ultraviolet ray lamp, a PL in red occurs.

Next, hydrogen ions accelerated at 30 keV were implanted into the surface of the water with a dose of 1 x 10% cm⁻² and subjected to RTO to effect activation and porous structure stabilization. The PL luminous peak was somewhat blue-shifted and increased in intensity.

After it had been confirmed that the porous layer oxidized to have the increased PL efficiency is in a sufficiently stabilized state, the oxide film on the top surface of this procus layer was removed with a diute equeeus "hydrofluoric acid solution. Before reformation of #; analura oldes film. TIO was vapor-deposited to a thickness of 1500A and then patterned in 5 mm square in the fashion of islands. An AISI film was deposited on the entire back surface of the water by southerino.

When a DC current was conducted between the electrodes on the tront and the back surfaces of the wafer of the device produced by the above steps, a rectification characteristic was indicated, where the current with the front surface electrode being the positive electrode is the forward direction, and a visible area luminescence of nearly orange color occurred from the ITO island portions at a threshold voltage of about 5 volts. It is considered that the light emission is an electroluminescence due to a current injection to the porous silicon layer with a ultra-fine structure via the junction interface between the p-type monocrystalline layer of the substrate and the porous layer as provided just on the p-type layer and changed to n-type by ectivation of hydrogen ions. It is presumed that the luminous threshold voltage being low sufficiently for practical use results from that the continuous crystal structure in the pnjunction interface provides good rectification characteristic and the current injection efficiency is higher than the conventional one.

(Example 5

An explanation will be made as for an example of a light emitting device having the structure shown in Fig. 2B manufactured in the process shown in Fig. 7.

A device was prepared by following the procedure of Example 4 with the exception that the accelerating energy for the hydrogen ion implentation was boosted to 45 keV and the dose amount was doubled. Although it is difficult to clearly other-mine by a cross section observation, it is presumed that the rybpe region extends to the Inside of the monocrystalline substrate through the porous layer.

When a DC current was conducted between the electrodes on the front and back surfaces of the wafer, the device indicated a rectification ratio being higher by nearly one order than the device of Example 4. The threshold voltage increased slightly to about 6 voltage that the Example 4. It is presumed that the improved rectification characteristic results from that the troit and back rections of the on-full-door interface is

formed of a perfect monocrystel and that this improves the efficiency of current injection into the luminous porous leyer, thus increasing the luminous efficiency.

(Example 6)

An explanation will be made as for an example of a light emitting device having the structure shown in Fig. 2C manufactured in the step shown in Figs. 9A to 9C.

A device was prepared by following the procedure of Example 1 with the exception that after a p type epitaxial layer had been grown on an n-type monocrystalline substrate, boron ions were implanted into the surface thereof at an accelerating voltage of 20 keV with a dose of 5 x 1014 cm-2 and the substrate was subjected to thermal annealing in a nitrogen atmosphere at 950 °C for 30 minutes to be activated. An observation of the cross section of a sample found that the anodization advances from the wafer surface up to about 1 µm depth to form a porous layer. In further deteil observation, it was recognized that the surface was of e porous layer having large pores and e thickness of about 0.05 um and the underlying lower porous layer was almost of a ultra-fine porous layer.

When a DC current was conducted between the front and the back surfaces of the wafer, both the rectification characteristic and the turninous threshold voltage were substantially the same as those in Example 1 and the furninous efficiency was improved several times that in Example 1. In this case, it is considered that since the porous layer with large pore structure on the outer surface acts as the injection electrode layer 3 shown in Fig. 2C, the contact resistance with the extracting electrode is reduced so that the luminous efficiency is improved.

(Evample 7)

An explanation will be made as for an light emitting device with the structure shown in Fig. 2C manufactured in the steps in Figs. 10A to 10D.

A device was prepared by following the procedure of Example 1 with the exception that the following step was added between the RTO treatment to the procus layer and the CVD-SOL, layer of deposition step. The added step includes removing the cride line on the top surface of the procus layer with the cridical surface of the procus sort with the cridical surface of the procus con layer of a 20th Library string. The proserved is a surface of the procusion of the critical surface of the critical surface of the surface of the critical surface of the critical wins and disclosulons were observed in the option. ial silicon layer. However, it was confirmed that the crystal structure of the underlying porous layer was succeeded as an average structure.

On conducting § DC current between the front and the back surfaces of the water, it was indicated that the rectification characteristic and luminous threshold voltage of the device was substantially the same as those in Example 1 and the luminous efficiency was improved several times that in Example 1. In this case, it is considered that since the oppiaxial silicon layer on the surface acts as the injection electrode layer 3 shown in Fig. 2C the contact resistance with the extracting electrode is reduced so that the luminous efficiency is improved.

(Example 8

An explanation will be made below as for a light emitting device with the structure shown in Fig. 2C formed in the steps shown in Fig. 11.

A device was prepared by following the procedure of Example 6 with the exception that the following step was added after the porous layer forming step and the RTO treating step. The added step is a step of, after an oxide film had been removed, epitaxially growing, a p-type silicon layer of about 150 Å thick by using a bias sputtering method where a p-type silicon with a high concentration was used as a target. In an observation of the cross section of a sample, an ultra-thin layer of about 0.05 µm in depth in the top surface side of the porous layer was converted to a porous structure with coarse pores. It may be said that the epitaxial layer on the porous layer is nearly a monocrystal because a defect or the like was not found therein. It is considered that the improved crystallinity of the epitaxial layer results from that the epitaxial growth surface is a porous with coarse pores, compared with that in Example 7.

On conducting a direct current between the front and back strates of the water of the above device, the rectification characteristics of the direct was substantially same as those in Example 6. The Luminous threshold voltage was slightly in creased, whereby the luminous efficiency was timported by about 50%. In this case, since the populated layer on the lop surface as well as the height of the condition of the control of the control of the control of the control of the condition of the condition of the control of the condition of the condition

(Example 9)

The example of the light emitting device with the structure shown in Fig. 2C formed in the steps

shown in Figs. 12A to 12D will be explained below

The same steps as in Example 4 was carried out in except that after the hydrogen los implaning step, the porous layer was subjected to a RTO treatment. After the coulde film on the top surface of the porous layer had been removed, an ehype sistion layer of about 200 Å ave splitakilly grown by using a bias sputtering method in which a n-type silicon with a high concentration was used as a target. In an observation of the cross section of an observation sample, some detects including disclosations and twins in the epitaxist silicon layer were found. However, it was confirmed that the crystal structure was continued from the underlying propous layer as a whole.

After it had been confirmed that the procus layer oxidized and with an increased Pt efficiency is sufficiently stabilized, about 2000Å thick SiO₂ film was again deposted on the surface of the water by using a CVD process. Using a conventional photoithography process, some openings of 2 x 2 mm² were formed in the oxide film. Moreover, a semi-transperent gold film of a thickness of 100Å was vapor-deposited in the openings to were the take-out electrodes AI derived from the open.

When a direct current was conducted between the electrodes on the back surface of the above water, the device produced by the above steps indicated a rectification characteristic in which a forward current flows to the surface electrode acting as a cathode electrode. An area light emission of nearly grange visible rays was emitted from the openings in the oxide film at a threshold voltage of about 2 volts. It is considered that this light emission is due to an electroluminescence based on a current injection to the porous silicon layer with the ultra-fine structure via the junction interface between the n-type epitaxial crystal layer and the underlying p-type porous layer. It is presumed that the sufficiently low luminous threshold voltage for practical use results from that the pn-junction interface has a continuous crystal structure and a good rectification characteristics and therefore the current injection elliciency is very high.

(Example 10

An explanation will be made as for an example of a light emitting device having the structure shown in Fig. 2D produced in the forming steps shown in Fig. 13.

The device was prepared in the same manner as in the Example 9 except that before converting the surface of a p-type monocrystalline substrate into a porous structure, boron ions with a dose of 5 x 10¹¹ cm⁻² were implanted to the surface thereous at an accelerating voltage of 20 keV. There the implanted surface was activated by annealing in a nitrogen amnosphere at 850° C for 30 minutes. An observation of the cross section of a sample found that the ultra-thin layely of about 0.05 µm in the top surface side of the portius layer was in a portous state with bulky strottyre and that the optisatial layer thereon included nearly no detects, and it can be stated that it is nearly a monocrystal. It is considered that the optisatial layer hardon includer noally no detects, and it can be stated that it is nearly a monocrystal. It is considered that the optisatial layer harmy a crystal-naying a crystal-naying activation.

linity better than that in the Example 7 results from

the epitaxially grown surface with bulky porous

On conducting a direct current between the electrodes on the front and back surfaces of the water, the rectification characteristics and luminous threshold voltage of the device over substantially the same as those in the Example 9 and the luminous efficiency was improved several times, compared with that in the Example 9. In this case, it is considered that the crystalling of the niyelf explantial silicon layer in the top surface reduced the contact resistance of the invertices to that the third product is the contact resistance of the invertices to that the third product is the contact resistance of the invertices to that the third product is the contact resistance of the invertices to that the third product is the contact resistance of the invertices to that the third product is the contact resistance of the invertices to that the third product is the contact resistance of the invertices to that the third product is the contact resistance of the invertices to the third product in the contact resistance of the invertices to the third product in the contact resistance of the invertices to the third product in the contact resistance of the invertices to the third product the invertices the contact resistance of the invertices to the third product the contact resistance of the invertices that the contact resistance is the contact resistance that the contact resistance

(Example 11)

An explanation will be made as for an Example of a light emitting device having the structure shown in Fig. 3D explained in the torming steps of Figs. 14A to 14F.

First, a SOI substrate having a boron-doped ptype silicon monocrystalline thin film with a face orientation of (100) and a thickness of 0.5 um formed on a transparent quartz substrate was provided. The SOI substrate was subjected to the LOCOS process to form 10 x 10 µm2 element separation regions. The surfaces of the separated silicon islands were exidized to about 500Å. Then with a stripe of 1.5 µm width being left at the middle portion of the island, boron ions accelerated at 150 keV were implanted with a dose of 2 x 1015 cm-2 into the right region and phosphorus ions accelerated at 150 keV were implanted with a dose of 3 x 1015 cm⁻² into the left region. The substrate was subjected to a thermal annealing in a nitrogen atmosphere at 950 °C for 30 minutes to activate it. Next, aluminum wirings were tormed in contact with the right and left ion implanted regions. Particularly, the wiring from the boron implanted region was wound to the orientation flat of the substrate. Then a conventional photoresist was coated on the surface of the substrate. An opening was termed in the non lon-implanted strine region left at the middle area of the silicon island at the surface side. The end of the opening at the pn-junction side was offset so as to overlap slightly the inner side of the stripe region from the junction boundary. Then the oxide tilm on the surface in the opening was removed.

The substrate was subjected to an anodizing process by immersing in a hydrofluoric acidethanol solution of a concentration ot 25 wt.% while a direct current was conducted between a platinum plate electrode acting as a cathode facing the substrate and the aluminum wirings winding to the orientation flat of the substrate. The anodizing process was completed in 30 seconds while the anodizing current density was controlled at a fixed value of about 20 mA+cm-2 at the photoresist opening on the surface of the substrate. Then the photoresist was removed from the substrate. A SiNx film was deposited on the entire surface of the substrate by using a plasma CVD process. An opening was formed at a portion where the aluminum wiring was needed.

An observation of the cross section of the device found that the 1.4 µm width stripe region at the middle portion of the silicon island became a porous structure. It is probably seemed that a predetermined intermediate layer with a width of less than about 0.1 µm is tormed at the pn-junction boundary.

On conducting a direct current between two electrodes sandwiching the porous region in the device, the device indicated its good rectification characteristics. A luminescence was recognized trom a small applied voltage of about 1 volt.

(Evample 12

An explanation will be made below as for an Example of a light emitting device having the structure shown in Fig. 3D produced in accordance with the steps in Figs. 5A to 5C. First, an SIMOX substrate having a phospho-

rus-doped n-type silicon monocrystalline thin film with a face orientation of (100), a resistivity of 20 Q+cm, and a thickness of 0.5 µm tormed on a buried oxide film of 3000 Å thick was provided. The SIMOX substrate was immersed in a hydrofluoric acid solution of a concentration of 49 wt-%. A photo-formation was carried out by irradiating a He-Ne taser beam focused in a rectangular shape of 2 x 8 µm² onto a part of the substrate surface for 30 minutes. Next, the substrate was subjected to the RTO process to stabilize the porous region. According to the same steps as those in the Example 11, an ion implantation was performed to the two regions sandwiching the porous region to isolate electrically them. The element separation step using the LOCOS process activates the implanted ions. In this case, the phosphorus ion-implanted region was spaced from the porous region by 0.1 μm and the boron ion-implanted region was spaced from the porous region by 0.2 um. Thus, the horizontal diffusion during an activation annealing was ceased at or just before the boundary of the porous region. An intermediate layer was formed between the boron implanted region and the porous region. Finally, aluminum wirings were derived out of both the ion implanted regions so that a device was conspletely formed.

23

On conducting a direct current between two electrodes sandwiching the porous region in the device, it was indicated that this device has good rectification characteristics, it was ensured that a light emission occurred at a small applied voltage of about 1.5 volts.

(Example 13)

An explanation will be made below as for an example of driving a light emitting device drive by an electrical circuit which is, fabricated on the same substrate including the light emitting device shown in the Example 11, with relerence to Figs. 16A to 16E and 17A to 17D.

First, a SOI substrate having a boron-doped ptype silicon monocrystalline film 200 with a face orientation of (100), a resistivity of 10 0 cm, and a thickness of 0.5 µm formed on an insulation surlace of a silicon substrate 0 was provided (Fig. 16A). The substrate was subjected to an element isolation by using the LOCOS process to form island regions 20 each having an area of 10 x 10 µm2. A 500A thick oxide film 7 was formed on the surface of each isolated silicon island by using a thermal oxidizing process (Fig. 16B). Next, a polycrystalline silicon film of 0.5µm thick was deposited on the surface of the oxide film 7 by using the LPCVD process. The polycrystalline siticon film was patterned through a conventional photolithography process so as to leave the 2 um wide island region 8 (Fig. 16C). Then, with the photoresist acting as a patterned mask, phosphorus ions with a dose of 2 x 1015 cm⁻² accelerated at 180 keV were locally implanted to the potycrystattine siticon film island region 8 as well as the parts 10 and 10' of the silicon monocrystalline film island region 20 at the both sides of the region 8. Moreover, boron ions with a dose of 1 x f0¹⁵ cm⁻² accelerated at 100 keV were implanted to a part 30 of the silicon monocrystalline film island region 20 by using the above step (Fig. 16D). The impurities were activated by a thermal process at 900 °C for 30 minutes. The oxide films on the surfaces of the regions 30 and 10' were removed, and then aluminum wirings 5 and 5' were taken out of the openings (Fig. 16E). Then the silicon oxide tilm 9 was deposited on the surface of the resultant substrate and the aluminum wiring 50 was taken out to conduct to the silicon monocrystalline film island region 9 (Fig. 17A). Next, the opening 100 was formed on a part of the upper part of the silicon monocrystalline film island 20 (Fig. 178). The silicon nitride litm 6 was tormed so as to leave an opening of 1.5um wide with a slight offset to the region 10 (Fig. 17C). Finally, the region 2 opened in the same way as that in the Example 11 was converted to a porous layer.

A light emitting device which includes the regions 10 and 30 corresponding to the ion-implanted electrode regions 1 and 3 shown in Fig. 14F. the region 2 corresponding to the perous region 2 including a light emitting region, and the offset region corresponding to the intermediate region 2' was formed according to the above steps (Fig. 17D). The ion-implanted electrode region 1 acts as the drain region of a MOS transistor including the region 10' acting as a source part, the region 20 acting as a channel part, and the polycrystalline silicon film island region 9 acting as a gate electrode. With a voltage of 15 volts applied between the wirings 5 and 5', when the gate voltage was increased from 0 via the wiring 50, the MOS transistor was turned on at a threshold voltage 1.1 volts. It was confirmed that the porous region including a light emitting region glowed at 1.5 volts and more. The relationship between the luminous intensity and gate voltage indicated linear characteristics over 2 volts. When a rectangular wavelorm of 5 volts was applied to the gate of a MOS transistor from a shift register separately formed on the same substrate, the light emitting device can follow even at 20 MHz.

The following features can be understood from

the above exptanations. A light emitting device according to the present invention has a device constitution comprising a non-porous region having a continuous crystal structure adjacent to the luminous region including a porous material and acting as a current injection electrode to the luminous region, whereby the perfect adhesive property between the electrode and the luminous region reduces the contact resistance. By differentiating the conductive types of the nonporous region and the luminous region from each other, a current injection from the junction portion could be injected with high elficiency. As a result, a light emitting device with excellent luminous efficiency and lor a practical use can be provided as a whole device. According to the present invention the light emitting device manufacturing method can provide light emitting devices with good luminous efficiency.

A tight emitting device comprises a luminous region comprising a luminous porous material comprising a crystalline semiconductor and a non-porous region adjacent to the luminous region,

wherein a conductive type between both the regions is ditterent at an interlace between the luminous region and the non-porous region and the tinuous.

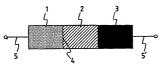
Claims

- A light emitting defice comprising a luminous region comprising a luminous porous material comprising a crystalline semiconductor and a non-porous region adjacent to the luminous region,
 - wherein a conductive type between both the regions is different at an interface between the luminous region and the non-porous region and the crystal structure between both the regions is continuous.
- A light emitting device according to claim 1, wherein a low resistance region having the same conductive type as that of the luminous region is further adjacent to the luminous region with a crystal structure being continuous.
- A light emitting device according to claim 1, wherein the luminous region comprises a porous region comprising plural porous layers which are different in composition and struc-
- A light emitting device according to claim 1, wherein the interface between the light emitting region and the non-porous region comprises a non-porous material.
- A light emitting device according to claim 1, wherein the interface between the luminous region and the non-porous region comprises a porous material.
- A light emitting device according to any one of claims 1 to 5, wherein the non-porous material is monocrystalline.
- A light emitting device according to any one of claims 1 to 6, wherein a heterogeneous material is formed on the porous material.
- A method of manufacturing a light emitting device comprising the steps of:
 - forming a member having a non-porous crystal region and a protus crystal region having a conductive type different from that of the non-porous crystal region, so as to continue a crystal structure of the porous crystal region and a crystal structure of the non-porous crystal region; and
 - forming a luminous region in the porous crystal region.

- A method of manufacturing a light emitting device comprising the steps of:
- lorming a member having a non-porous crystal region and a porous crystal region, so as to continue a crystal structure of the porous crystal region and a crystal structure of the non-porous crystal region;
- making a conductive type of the porous crystal region different from that of the non
 - porous crystal region; and forming a luminous region in the porous
 - forming a luminous region in the porous crystal region.
- 10. A method of manufacturing a light emitting device according to claim 8, wherein the porous region is formed by anodizing a crystal region formed by epitaxial growth.









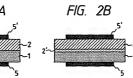
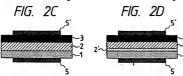
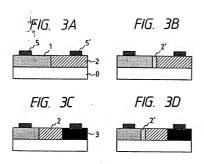


FIG. 2C





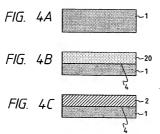






FIG. 5B



FIG. 5C

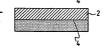


FIG. 6

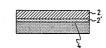
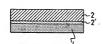
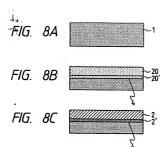
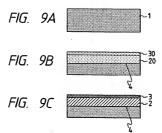


FIG. 7







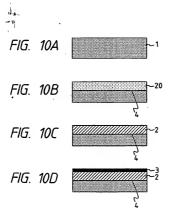




FIG. 12A

F.IG. 12B

FIG. 12C

FIG. 12D

FIG. 13

FIG. 14A



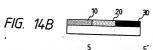








FIG. 14F



